

Structural stability of multiferroics BiMnO₃ under high pressure

Z. H. Chi · S. J. You · L. X. Yang · L. C. Chen ·
C. Q. Jin · X. H. Wang · R. Z. Chen · L. T. Li · Y. C. Li ·
X. D. Li · J. Liu

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Abstract Multiferroics BiMnO₃ was fabricated via high pressure of 4 GPa. The crystal structure of as-prepared specimen was determined by x-ray diffraction to be highly distorted monoclinic perovskite with space group of C2. Multiferroism (or ferroelectromagnetism, i.e., coexistence of ferromagnetism and ferroelectricity) in as-prepared specimen has been reported in the recently published paper (Z.H. Chi, C.J. Xiao, S.M. Feng, F.Y. Li, C.Q. Jin, J. Appl. Phys. 98:103519, 2005). Structural toughness of as-prepared specimen under high pressure was studied in diamond-anvil cell (DAC) combined with synchrotron radiation x-ray diffraction. No structural phase transition was detected up to the maximum pressure of 27 GPa, indicating a compact and low compressibility of monoclinically distorted perovskite structure of BiMnO₃. The bulk modulus $B_0=141$ GPa ($B'_0=4$) was derived from Birch–Murnaghan equation of state (EOS).

Keywords Multiferroics · DAC · Synchrotron radiation · Bulk modulus · Equation of state

Z. H. Chi · S. J. You · L. X. Yang · L. C. Chen · C. Q. Jin (✉)
Institute of Physics, Chinese Academy of Sciences,
Beijing 100080, People's Republic of China
e-mail: cqjin@aphy.iphy.ac.cn

X. H. Wang · R. Z. Chen · L. T. Li
State Key Lab of New Ceramics and Fine Processing,
Department of Materials Science and Engineering,
Tsinghua University,
Beijing 100084, People's Republic of China

Y. C. Li · X. D. Li · J. Liu
Institute of High Energy Physics,
Beijing High Pressure Research Center,
Chinese Academy of Sciences,
Beijing 100039, People's Republic of China

1 Introduction

Magnetolectric multiferroics, sometimes termed as ferro-electromagnet, exhibiting ferroelectricity and ferromagnetism simultaneously, i.e., being both magnetically and electrically polarized, has stimulated a great deal of research interest during the very recent years [2–15]. Apart from the tremendous application perspective based on the mutual control of polarization and magnetization via magnetolectric effect, i.e., the induction of electric polarization by means of magnetic field or vice versa, the fundamental physics behind is fascinating. Unfortunately, the number of ferroelectromagnet is dramatically reduced to very few cases with respect to pure ferroelectrics or ferromagnetics due to the incompatibility between magnetism and ferroelectricity in both electronic and crystal structure requirements. It is well established that magnetism and ferroelectricity are involved with local spins and off-center structural distortion, respectively. Through first principles calculation, Hill et al. [6] has pointed out that partially occupied orbital of transition metal *d* electron which is prerequisite for magnetism reduces the tendency for off-center structural distortion which is the origin of ferroelectricity. Thus, an additional structural driving force must be present to engender the concurrence of ferroelectricity and ferromagnetism. Lone pair of electrons has proven to play a pivotal rule in inducing structural distortion and ferroelectricity in typical ferroelectric PbTiO₃ [7], which exhibits an entirely different polarization mechanism in comparison with homologous BaTiO₃ due to the stereochemical activity of lone pair on Pb²⁺ cation. Homologous candidate ferroelectric PbVO₃ [8] with gigantic spontaneous polarization surpassing that of PbTiO₃ as a result of larger tetragonality has been fabricated by high pressure–high temperature synthesis technique. Like Pb²⁺, Bi³⁺ with 6s² lone pair is the key factor in triggering

ferroelectricity in perovskite multiferroics BiFeO_3 [9]. Very recently, BiMnO_3 was unraveled to be a new supplement to the scarce multiferroics family, in which magnetocapacitance effect (MC) has been verified [10]. Due to the fairly strong covalent character of $6s^2$ lone pair, BiMnO_3 can only be synthesized at high pressure of at least 3 GPa [11] to stabilize the highly distorted perovskite structure. Once quenched from high temperature–high pressure state, it survives as a metastable phase with $C2$ monoclinic structure at ambient condition [12]. Multiferroism (or ferroelectromagnetism) with ferroelectric Curie temperature T_E at 773 K and ferromagnetic Curie temperature T_M at 105 K in BiMnO_3 has been widely recognized. So far, investigation on structural feature of BiMnO_3 has been focused on temperature dependence only [13]. Structural evolution of BiMnO_3 when subjected to external pressure still remains open question. In this paper, we investigated the stability of non-centrosymmetric polar structure of BiMnO_3 under high pressure up to nearly 27 GPa.

2 Experimental

Polycrystalline BiMnO_3 ceramic was fabricated via a high pressure–high temperature synthesis route. Stoichiometric mixture of Bi_2O_3 (Alfa Aesar, 99.99%) and Mn_2O_3 (Alfa Aesar, 98%) reagents was intimately ground, pelletized and enveloped with gold foil to circumvent contamination from the ambience. The ultimate synthesis was conducted in a cubic anvil-type apparatus by keeping the specimen at 1,273 K and 4 GPa for 30 min. Pressure was released slowly after quenching the specimen to room temperature.

Angle-resolved powder x-ray diffraction pattern was recorded with $\text{Cu K}\alpha$ radiation ($\lambda=1.54050 \text{ \AA}$) at M18AHF diffractometer (MAC SCIENCE, JAPAN). Data processing and structure refinement were performed employing *Powder X* software packet [14].

High-pressure synchrotron x-ray diffraction experiment was performed on 3W1A high-pressure beamline at Beijing Synchrotron Radiation Facility (BSRF). The diffraction pattern was recorded with white synchrotron radiation in an energy-dispersive mode. Finely ground powder of BiMnO_3 together with a tiny piece of ruby crystal for calibrating the inner pressure was loaded in a minute hole with diameter of 250 μm , which serving as sample chamber was drilled in the gasket made of T301 stainless steel. Then mixture of methanol and ethanol (4:1) as pressure-transmitting medium was injected into the chamber to ensure better quasi-hydrostatic pressure condition. The inner pressure in the chamber was monitored via the shift of the fluorescence bands of Cr^{3+} in the ruby crystal. Polychromatic radiation was collimated to a spot with size of $30 \times 20 \mu\text{m}^2$ in area. High pressure was generated in DAC (diamond-anvil cell)

with diamond culet size of 500 μm . Germanium detector was located at a fixed angle of 7.45° with respect to the incident synchrotron radiation. The d spacing of different reflection ($h k l$) of BiMnO_3 specimen was calculated according to the Bragg law expressed as following:

$$E * d = \frac{6.19925}{\sin \theta} \quad (1)$$

3 Results and discussion

Figure 1 exhibits the angle-resolved powder XRD pattern of as-prepared specimen recorded at room temperature. Impurities including $\text{Bi}_2\text{O}_2\text{CO}_3$ and A-type Bi_2O_3 were detected despite enormous efforts made by changing stoichiometry of starting oxides, reaction pressure, reaction temperature and reaction time to improve the purity. The main peaks in the pattern can be indexed on the basis of $C2$ monoclinic structure exclusive of peaks from impurities, in good accordance with the conclusion of neutron and electron diffraction [12].

In situ high-pressure energy-dispersive synchrotron X-ray diffraction spectra of BiMnO_3 are displayed in Fig. 2. The spectra were collected at pressures ranging between atmospheric and 26.8 GPa. Through deliberate analysis, nine diffraction peaks with corresponding indices of (111), (-113) , (-311) , (004), (-404) , (222), (-315) , (313) and (115) can be assigned to the $C2$ monoclinic phase of BiMnO_3 in spite of poor profile and intensity of peaks. It is very explicit that all the diffraction peaks of $C2$ phase of BiMnO_3 shift to the higher energy side with increasing pressure, which coincides well with the Bragg law (1). Neither new diffraction peaks emerge nor original peaks

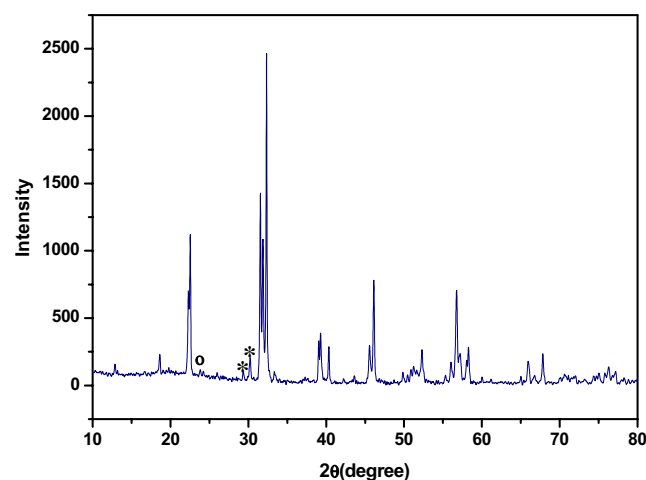
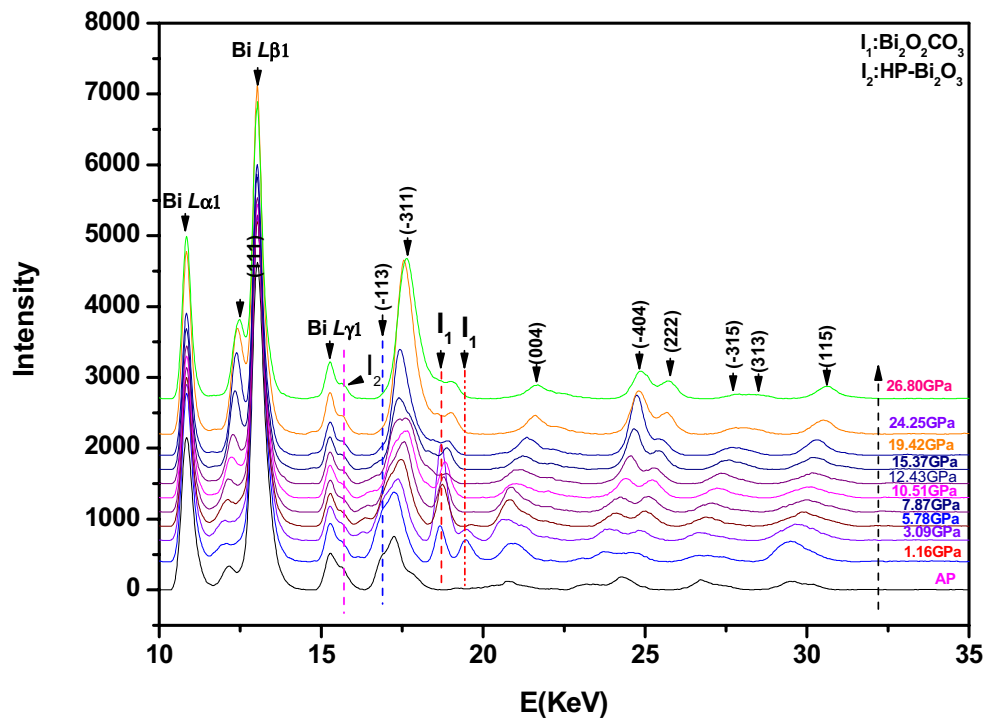


Fig. 1 Angle-resolution powder X-ray diffraction pattern of as-prepared BiMnO_3 ceramic recorded by $\text{Cu K}\alpha$ radiation at room temperature (*asterisk* and *open circle* denote A-type Bi_2O_3 and $\text{Bi}_2\text{O}_2\text{CO}_3$ impurity, respectively)

Fig. 2 In situ high-pressure energy-dispersive x-ray diffraction pattern of as-prepared BiMnO₃ ceramic recorded by white synchrotron radiation at room temperature (Bi L_{α1}, Bi L_{β1} and Bi L_{γ1} indicate the fluorescent peaks of element Bi, respectively)



disappear in the full pressure range, indicating no structural phase transition takes place during the compression process.

Lattice parameters of C2 phase of BiMnO₃ with increasing pressure were calculated by virtue of modification of *d* spacing of detectable diffraction peaks. The calculated lattice parameter was plotted as a function of pressure in Fig. 3. As shown in the plot, lattice parameters along three fundamental axes change not monotonically but with sizable fluctuation, this can be attributed to the non isotropic compression as a result of poor hydrostatic condition.

In order to obtain an evaluation of the compressibility, volume compression ratio of V/V_0 as a function of pressure

was plotted in Fig. 4. The scattered dot was fitted nonlinearly with Birch–Murnaghan equation of state (EOS):

$$P(\text{GPa}) = \frac{3}{2} * B_0 * \left[(V_0/V)^{7/3} - (V_0/V)^{5/3} \right] * \left\{ 1 - \left(3 - 3 * B'_0/4 \right) * \left[(V_0/V)^{2/3} - 1 \right] \right\} \quad (2)$$

Supposing the first-order derivative $B'_0 = 4$, we obtain the bulk modulus $B_0 = 141 \pm 4$ GPa, which is comparable to most oxide ceramics.

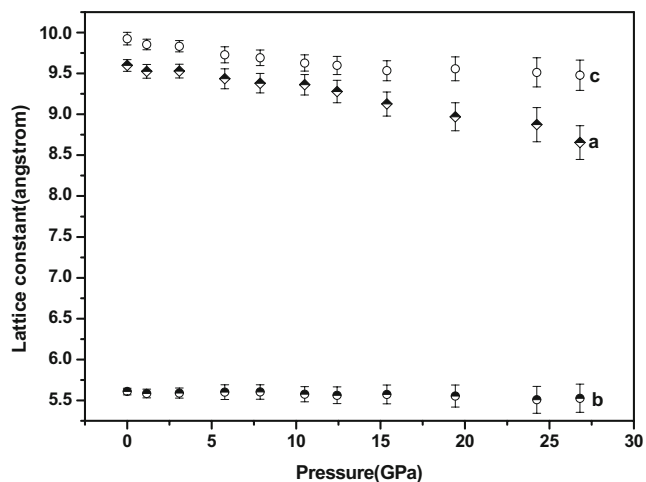


Fig. 3 Pressure dependence of lattice parameters of as-prepared BiMnO₃ ceramic

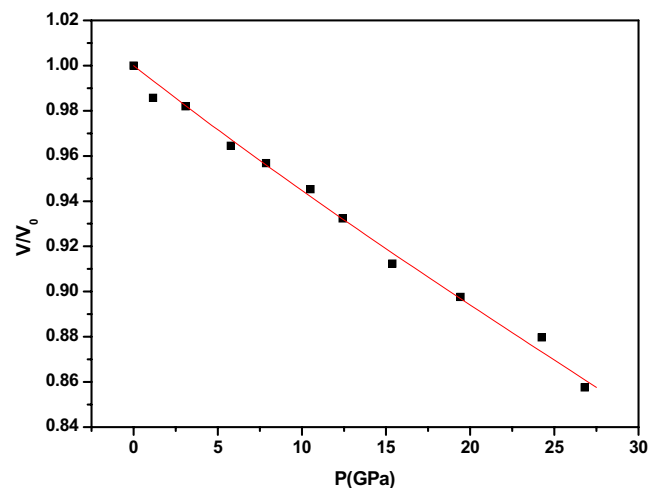


Fig. 4 Pressure dependence of volume compression ratio of as-prepared BiMnO₃ ceramic

4 Conclusion

Multiferroics BiMnO₃ ceramic with a highly distorted perovskite structure was synthesized and studied under high pressure. It is inferred that the *C2* monoclinic structure is stable up to nearly 27 GPa. Bulk modulus B_0 was calculated to be 141 GPa from Birch–Murnaghan equation of state.

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